Self-Heating in GaN Transistors Designed for High-Power Operation

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Abstract—DC and transient self-heating effects are investigated in normally off AlGaN/GaN transistors designed for a highpower operation. Electrical and optical methods are combined with thermal simulations; 2-µs-long voltage pulses dissipating about 4.5 W/mm are applied on four different transistor structures combining GaN or AlGaN buffer on an n-type SiC substrate with or without Ar implantation. Transistors with only 5% Al mass fraction in the buffer show almost a threefold increase in the transient self-heating if compared with devices on the GaN buffer. On the other hand, 2- μ s-long pulses were found not to be long enough for the Ar-implanted SiC substrate to influence the device self-heating unless AlGaN composition changes. In the dc mode, however, both the buffer composition and Ar implantation significantly influence the self-heating effect with the highest temperature rise for the transistor having the AlGaN buffer grown on the Ar-implanted SiC. We point on possible tradeoffs between the transistor high-power design and the device thermal resistance.

Index Terms—GaN, high-electron mobility transistor (HEMT), optical characterization, thermal characterization, thermal resistance.

I. INTRODUCTION

aN-BASED transistors are considered to be excellent candidates for high-power and efficient switching applications [1]. Various design issues were investigated in the past to boost the device performance. The usage of a Fe-doped GaN or AlGaN buffer instead of an ordinary undoped GaN was suggested to increase the device breakdown voltage while

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maintaining a low dynamical on-resistance [2]. Similarly, Ar implantation in the n-type SiC substrate was shown to improve a vertical blocking voltage in GaN high-electron mobility transistor (HEMT) [3], [4]. However, no attention was particularly given to investigate an impact of the mentioned or similar design changes on the device self-heating. Elevated temperature during the operation of GaN-based transistor may deteriorate the device performance [5] and reliability [6]. In this paper, we investigate dc and transient self-heating effects in tailored designed high-power normally off GaN HEMTs using electrical [7], [8] and optical [9] methods.

II. DEVICES AND METHODS

Investigated AlGaN/GaN/AlGaN HEMT devices use a p-type gate technology [10]. All devices were metal-organic chemical vapor deposition grown on the n-type SiC, with or without Ar implantation. The epi-structures comprise AlN seeding layer, AlGaN buffer with targeted 5% Al mass fraction or alternatively a sequence of Fe-doped and undoped GaN, AlGaN barrier layer, and p-type GaN (Table I). Combinations of different buffers and/or implantation define structures A-D. Structures were depleted from electrons due to p-GaN. As reported elsewhere, after removing p-GaN layer, Hall-measurement data indicated about 10% drop in electron concentration data in heterostructures with AlGaN buffer if compared with GaN ones, from 7.27×10^{12} down to 6.58×10^{12} cm⁻², with only minor change in mobility from 1615 down to 1535 cm²/Vs [4]. With Ar implantation, changes were between -1% and -5% [4]. Lateral device geometrical dimensions were gate length $L_G = 1.3 \mu m$, gate width $w = 2 \times 1.05$ mm, gate-to-drain distance $d_{GD} = 15 \mu m$, and source-to-gate distance $d_{SG} = 1 \mu m$. Wafers were diced to chips with a size of 1.8 mm \times 3.76 mm \times 0.36 mm and brazed on copper flanges. Other technological details can be found elsewhere [2], [3], [10].

Self-heating was monitored in the steady state and during 2- μ s-long voltage pulses (1.8 μ s after subtracting the rising edge). The dc steady-state method is based on the effect of the HEMT saturated drain current drop $\Delta I_{\rm sat}(V_D)$ as the drain voltage is increased. This effect was described analytically as [7]

$$\Delta I_{\text{sat}}(V_D) = -g_m(I_{\text{sat}} \Delta R_S + \Delta V_T) + I_{\text{sat}} \Delta v_{\text{sat}} / v_{\text{sat}} + V_D / R_{\text{out}}$$
 (1)

| Structure | SiC substrate | AIN thickness | Buffer type | Buffer thickness | Channel thickness | AlGaN barrier | p-GaN |
|-----------|------------------|------------------|--|---------------------|-------------------|------------------|--------|
| Α | n-type | 100 nm | AlGaN | 3560 nm | 35 nm | 14 nm 24 % | 103 nm |
| В | n-type + Ar | 100 nm | AlGaN | 3560 nm | 35 nm | 14 nm 24 % | 103 nm |
| С | n-type | 80 nm | 200 nm nidGaN + 2070 GaN:Fe + 880 nm nidGaN | | | 14 nm 23 % | 102 nm |
| D | n-type + Ar | 80 nm | 200 nm nidGaN + 2070 GaN:Fe + 880 nm nidGaN | | | 14 nm 23 % | 102 nm |

TABLE I

EPI-Structure Details of Studied Structures. Channel Thickness Is

Indicated Only for AlGaN Buffer Samples A and B

where g_m is the transconductance, ΔR_S , ΔV_T , and $\Delta v_{\rm sat}$ are temperature-driven changes in a source resistance, threshold voltage and electron saturation velocity, respectively. V_D represents the drain voltage, and $R_{\rm out}$ represents a differential output resistance. Calibrating steady-state dependencies of the transistor parameters on temperature (ΔR_S and $\Delta V_T = f(T)$, $\Delta v_{\rm sat}$ is neglected introducing about 10% overestimation in T), it is possible to determine the HEMT channel temperature as a function of dissipated power $V_D x I_{\rm sat}$.

To describe a temperature-induced *time-dependent* current drop $\Delta I_{\rm sat}(t)$, we use methodology of the dc method, considering several differences [8]. First of all, as the voltage is kept constant during pulses, the same is assumed for the buffer/drain modulation leakage current and the last term in (1) could be neglected. Similarly, for a relatively short duration of the pulse, we neglected trap-related ΔV_T . Therefore, $\Delta I_{\rm sat}(t)$ can be expressed as

$$\Delta I_{\text{sat}}(t) = -g_m I_{\text{sat}} \Delta R_S. \tag{2}$$

Using a polynomial fit to calibration curves $\Delta R_S = f(T)$, and $g_m(T) = g_m(300 \text{ K})/[1+g_m(300 \text{ K}) \times \Delta R_S(T)]$ together with the measured $\Delta I_{\text{sat}}(t)$, an iterative solution of (2) could be performed to obtain the channel temperature transient characteristics [8].

To determine ΔR_S at different T, initially, we determine $R_{S,D}=R_S+R_D$ by extrapolating a total source-to-drain resistance R_T dependence to $V_x=1/\{1-[(\Phi_b-V_{\rm GS})/(\Phi_b-V_T)]\}$ = 0 V [11], where $\Phi_b\sim 3.8$ V is given by a p-n junction built-in voltage and conduction band discontinuity at p-GaN/AlGaN interface. Then, R_S is calculated as $(R_{S,D}-2\times R_C)\times d_{\rm SG}/(d_{\rm SG}+d_{\rm GD})+R_C$, where $R_C=0.2$ Ω mm is the contact resistance.

In the transient interferometric mapping (TIM) method, the device is scanned from the backside using an infrared laser beam (3 ns, 1.5- μ m resolution). A phase shift $\Delta \varphi$ of a reflected beam (from the topside) is caused by a temperature-induced change in a material refractive index n along the beam path L and can be expressed as [12]

$$\Delta\varphi(x,y,t) = \frac{4\pi}{\lambda} \int_0^L \frac{dn}{dT} \Delta T(x,y,z,t) dz.$$
 (3)

Here, λ is a laser wavelength and x and y are lateral coordinates. The TIM method operates in the transient (pulsed) regime of the self-heated device and can be used to trace the lateral and vertical spreading of the heat [8], [9].

To simulate and explain self-heating effects in the transient state of the devices, we used a 2-D thermal model SDevice from Synopsys [13]. We used thermal conductivity values (at room temperature) $K_{\rm GaN}=150$ W/mK [14] and $K_{\rm SiC}=490$ W/mK [9], while $K_{\rm AlGaN}$ of the buffer layer was used as a fitting parameter. Substantial drop in $K_{\rm AlGaN}$ even for a low Al mass fraction in the buffer can be expected [14]. Similarly, decreased $K_{\rm SiC}$ can be expected in the top 300 nm of the implanted substrate where Ar penetrates [3]. Effective $K_{\rm AlN}=2$ W/mK was chosen to simulate a thermal boundary resistance at the substrate/nitride interface to be 5×10^{-8} m²K/W [9], [15]. Both $K_{\rm GaN}$ and $K_{\rm AlGaN}$ are assumed to be temperature dependent [14] with a factor T^{-1} .

III. RESULTS AND DISCUSSION

A. Device Calibration

Fig. 1(a)-(d) shows dc characteristics of the studied GaN-based normally off transistors. Slightly lower I_{sat} were observed for AlGaN buffer structures A and B, because of a depletion effect introduced by the back-barrier [11]. Note the pronounced negative differential output conductance in all studied transistors, particularly for structures A and B starting already at $I_{\rm sat} \sim 100$ mA/mm. This phenomenon indicates unusually strong self-heating effects even though SiC has been used as a substrate. Large total gate width combined with small chip dimensions and a limited heat removal to the flange can account for this effect. Thermal calibrations of R_S (starting from $\sim 1~\Omega$ mm at room temperature) up to 180 °C are documented in Fig. 2 as ΔR_S , including polynomial fits to the experimental data. Similarly, V_T dependences were fitted to describe the observed V_T rise between 150 mV for structure D and 500 mV for structure A, respectively (not shown). Fig. 3 shows the current and voltage waveforms during $2-\mu$ s-long voltage pulses applied on each transistor at different gate bias points to dissipate about 4.5 W/mm in average. About 200-ns-long rising edges including short RC-related humps were neglected by temperature extractions.

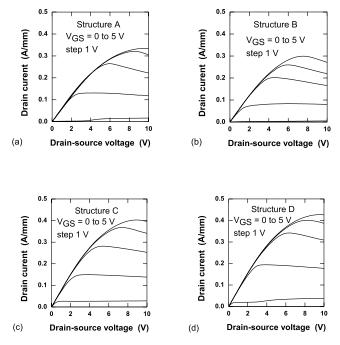


Fig. 1. DC output characteristics of structure (a) A, (b) B, (c) C and (d) D.

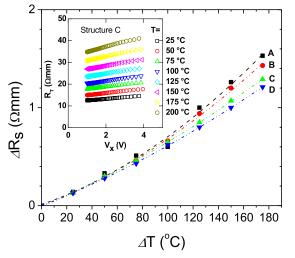


Fig. 2. Thermal calibration of the source resistance R_S of structures A–D. Inset shows source-to-drain resistance R_T of structure C measured at $V_D=100$ mV at different T as a function of $V_X=1/\{1-[(\Phi_b-V_{\rm GS})/(\Phi_b-V_T)]\}$.

B. Transient Self-Heating

Extracted T transients, including simulation curves, are shown in Fig. 4. Almost identical and the lowest self-heating effects were observed for structures C and D, while the effect becomes dramatically stronger for structures A and B. Best fits were observed with $K_{\rm AlGaN}=30$ W/mK for structure A and $K_{\rm AlGaN}=15$ W/mK for structure B. On the other hand, simulated transient self-heating effects were found to be practically invariant to the thermal conductivity value of Ar-implanted SiC for all structures (not shown). This analysis suggest that: 1) transient self-heating effect is dramatically increased by replacing GaN buffer with AlGaN and 2) $2-\mu$ s-long pulse is short enough to prevent a significant heat penetration to the substrate. Later finding is supported

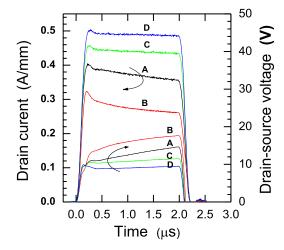


Fig. 3. Current-voltage waveforms of structures A–D during $2-\mu$ s-long pulses dissipating about 4.5 W/mm.

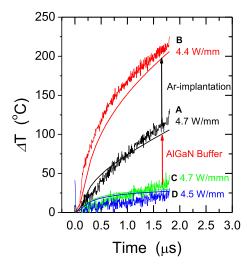


Fig. 4. Extracted (noisy lines) and calculated (full lines) temperature rises in structures A–D. In calculations, time evolutions of dissipated power levels are taken from waveforms shown in Fig. 3, and the average dissipated power values are marked close to each line. Arrows highlight the changes introduced to self-heating by different technological designs.

also by calculated T depth profiles at the end of pulses where negligible T rises in SiC can be inferred from Fig. 5. Thus, the most pronounced self-heating effect observed in Ar-implanted structure B can be related to lower $K_{\rm AlGaN}$ compared with that of structure A rather than to modified $K_{\rm SiC}$. We note that the AlGaN buffer is slightly thicker than the GaN buffer. Nevertheless, qualitative differences in self-heating between GaN and AlGaN buffer devices shown in Fig. 5 would be even more pronounced having the same buffer thicknesses because of the heat confinement imposed by the thermal boundary resistance at the buffer/SiC interface.

Our findings were further supported by TIM experiments showing similar signal evolutions for structures C and D, while there are clear differences between the phase shifts for structures A and B (Fig. 6). Moreover, TIM signals for structures A and B with AlGaN buffer are overall reduced if compared with structures C and D, with the lowest signal for structure B. This is because the thermo-optical parameter dn/dT of AlGaN is supposed to be lower than that of GaN [16].

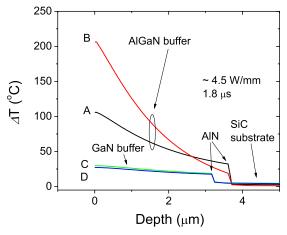
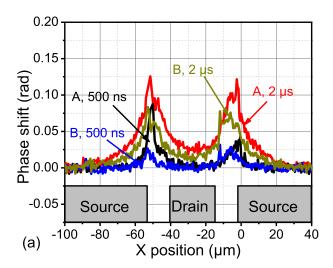


Fig. 5. Calculated temperature rises in the cross section of the devices at the end of 1.8- μ s-long pulses dissipating about 4.5 W/mm.



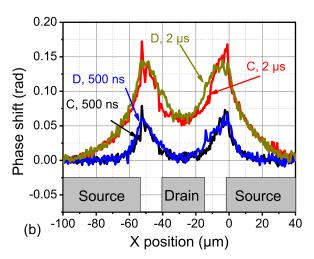


Fig. 6. TIM signals at 500 ns and 2 μs from the beginning of pulses for structures (a) A and B and (b) C and D.

Thus, for the short pulses, the heat spreading (both vertical and lateral) is not affected by Ar implantation in SiC alone (see comparison of TIM signals between structures C and D). Rather, higher Al content in the AlGaN buffer for structure B grown on Ar-implanted SiC can be assumed. Our finding

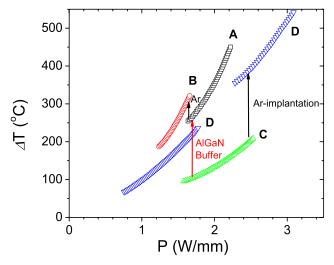


Fig. 7. DC temperature rises as a function of dissipated power for structures A–D. Arrows highlight the changes introduced to self-heating by different technological designs. For structure D, temperature rises measured both at $V_{\rm GS}=3$ V and $V_{\rm GS}=2$ V are shown.

also correlates with the reported intact structural quality of the buffer layer grown on Ar-implanted SiC [3], i.e., TIM difference between structures A and B may not be explained by degraded buffer. We speculate that changed properties of Ar-implanted SiC surface give different incorporation of Al during the growth and consequently increased Al mass fraction in the buffer. We note that even a slight variation in Al fraction between 0% and 10% may have a significant impact on $K_{\rm AlGaN}$ [14]. It was analyzed elsewhere that a small portion of alien atoms in host material produces profound effect on the phonon scattering [14]. Therefore, the thermal conductivity of the buffer of the structure B is minimal from all the studied samples with the most pronounced self-heating effect.

C. DC Self-Heating

In Fig. 7, we show extracted T values in the dc regime of devices as a function of dissipated power. Calculations were done using (1) for devices biased at $V_{GS} = 3$ V and $V_{\rm GS} = 2$ V (structure D only) and by extrapolating $\Delta R_S =$ f(T) polynomial fitting curves. We note that ΔR_S thermal calibration was done only up to 180 °C, and thus temperature extraction above this limit is an estimate. Nevertheless, similarly as observed in transient measurements in Fig. 4, the level of the self-heating is dominantly controlled by the buffer composition. Thus, the highest device thermal resistance is observed for structures A and B and statement 1) from the previous paragraph can also be adopted for the dc regime. On the other hand, the dc regime brings qualitative differences in a role of Ar implantation as the heat in the dc case is removed also through the substrate. As shown in Fig. 6, the impact of Ar implantation can be partially screened in the devices with the low thermal conductance of the AlGaN buffer; however, it is clearly visible from the comparison between structures C and D having GaN buffer with the high thermal conductivity. Therefore, 3) device cooling in the dc state is partially blocked by the degraded lattice of Ar-implanted SiC.

IV. CONCLUSION

In conclusion, we studied pulsed and dc regime self-heating effects in GaN transistors designed for high-power operation using electrical and optical methods and by thermal simulation. It is found that replacing the GaN buffer with AlGaN significantly increases transistor's self-heating. On the other hand, Ar implantation to the SiC substrate does not play a role in shortly pulsed transistors with GaN buffer because of the heat confinement in the buffer, while it overall limits the device cooling in the dc state. Our findings point on possible tradeoffs that must be given in to consideration by designing high-power GaN transistors.

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