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1. Introduction

The terahertz (THz) spectral region ranging between 30 μm – 300 μm covers various relevant and significant real-world applications like spectral imaging [1], medical diagnostics [2] and trace gas spectroscopy [3]. Unfortunately, due to the lack of high-performance laser sources, it is also referred to as the “THz-gap” in the electromagnetic spectrum. Therefore, a crucial step is to close this gap by realizing room temperature emitting THz laser sources.

To date, GaAs-based quantum cascade lasers (QCLs) [4] are the most promising devices but they lack significant improvements within recent years concerning their maximum operating temperature and are still limited to operation at cryogenic temperatures ($\sim 200\text{K}$). They are fundamentally limited by the parasitic, non-optical LO-phonon transitions (36meV in GaAs), being on the same order as the thermal energy at room temperature ($kT = 26\text{meV}$). Promising alternative semiconductors to solve this problem include new material systems like ZnO or GaN with their larger LO-phonon energy ($E_{\text{LO,ZnO}} = 72\text{meV}$, $E_{\text{LO,GaN}} = 91\text{meV}$). While GaN has already been extensively investigated in the past without final breakthrough concerning GaN-based QC devices, ZnO is new to this field of application.

To master the fabrication of ZnO-based QC structures, a high quality epitaxial growth is crucial combined with a well-controlled fabrication process including (selective) Zn(Mg)O etching, and the deposition of low resistance ohmic contacts.

2. Results

The devices we investigated are grown on m-plane [10-10] ZnO-substrate by molecular beam epitaxy (MBE). Their core typically consists of a ZnO quantum well, sandwiched between two ZnMgO barriers, while above and below highly doped ZnO-layers ensure good electrical contacts. They are patterned by (dry) plasma etching in a CH₄-based chemistry into square MESAs (see Fig. 2). The CH₄-process protects the mask by an amorphous carbon-layer increasing the selectivity of the etching [4].

Resonant tunneling diode structures are investigated in this geometry and are presented including different barrier- and well-configurations (see Fig. 3 for a typical RTD current-voltage characteristic). We extract contact resistances of $\sim \text{mid } 10^{-5} \Omega \text{ cm}^2$ to $\sim \text{mid } 10^{-4} \Omega \text{ cm}^2$ for the top and bottom contact, respectively, for not annealed Ti/Au contacts and an electron mobility of above 130cm²/Vs. Those values are in good agreement with typical values from literature. Demonstrating resonant electron tunneling in Zn(Mg)O structures is a crucial prerequisite for future QCL and QCD devices.

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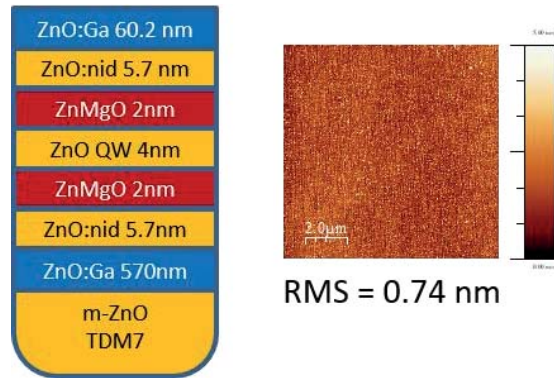


Fig. 1: (Left): Layer-sequence of a typical RTD-structure. (Right): AFM measurement of the same MBE-grown structure, showing the low surface roughness (RMS = 0.74 nm).

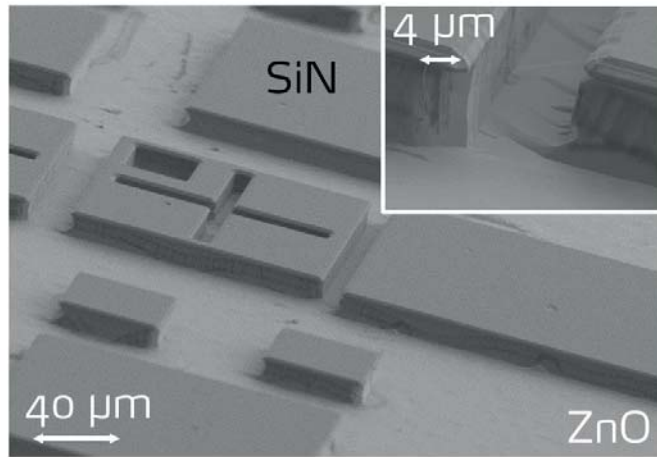


Fig. 2: Typical ZnO-based MESA RTD structures after ICP-RIE dry and additional H₃PO₄ wet etching. The inset shows the vertical and smooth sidewalls.

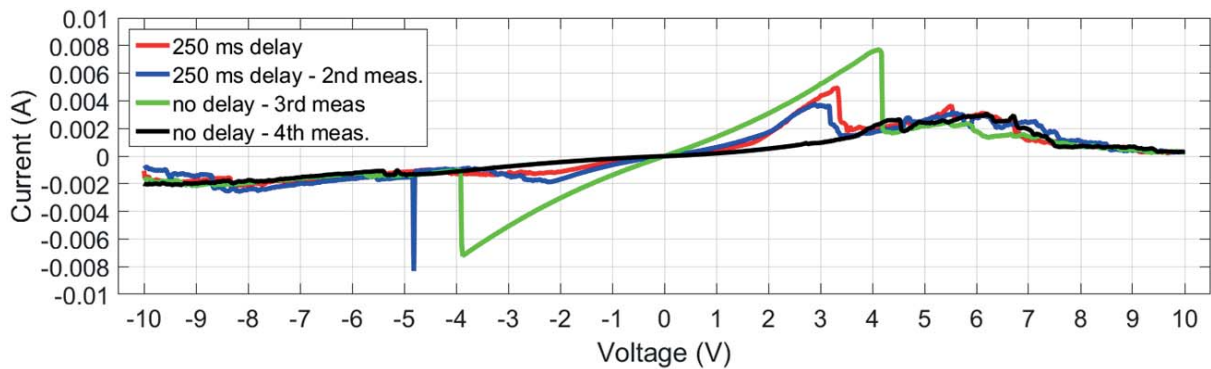


Fig. 3: Comparison of exemplary IV-curves of a 75 μm MESA RTD structure at room temperature with (250 ms) and without additional delay for each measurement point (resolution: 0.01 V).