

Article CO Oxidation Capabilities of La- and Nd-Based Perovskites

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Abstract: Catalytic tests to assess the performance of mixed perovskite-type oxides (La_{0.9}Ca_{0.1}FeO_{3- δ}, La_{0.6}Ca_{0.4}FeO_{3- δ}, Nd_{0.9}Ca_{0.1}FeO_{3- δ}, Nd_{0.6}Ca_{0.4}FeO_{3- δ}, Nd_{0.6}C

Keywords: CO oxidation; rare earth perovskites; heterogeneous catalysis; catalyst design



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1. Introduction

Oxide-type perovskites are a material class composed of two cations A and B (with the general formula ABO₃), where A is larger than B. Because of these fairly general criteria, perovskites are extremely versatile materials—provided the size ratio of A and B obey the well-known Goldschmidt stability factor [1], almost all elements across the periodic table can be used to make up the basic structure. Additionally, doping—both on A- and B-sites—is relatively easy and allows systematic tuning of the properties towards a given application [2,3]. Because of this compositional flexibility and their varied properties, perovskites are already used in a wide range of applications, for example in catalysis, fuel cells, or solid-state electrochemistry [4–10]. Properties that make perovskites interesting materials for use in industrial settings are in particular their thermal stability (and, consequently, the possibility to use them at high operating temperatures) as well as their resistance against catalyst poisons in connection with their regeneration capabilities via redox cycling.

In the last decades, La-based perovskites have been a topic of interest in catalyst research in biofuel production. For example, studies report utilization of LaMeO₃ (Me = Sr, Fe, Co) in Fischer–Tropsch syntheses as support for CoO clusters [11] or directly as catalysts [12–14]. Similar La-based materials have also been used in catalytic synthesis of alcohols [15,16], and Cu-doped titanates have been recently investigated as catalysts for CO₂ hydrogenation via computer simulations [17].

CO oxidation is widely used as a—seemingly—simple test reaction to assess catalytic performance of heterogeneous catalysts, especially for other oxidation processes [18]. Perovskites as catalysts for CO oxidation were mentioned 50 years ago as alternatives to precious metal catalysts in three-way catalysts, serving as an example of a technological application of CO oxidation [19,20]. Since then, extensive research into a broad range of different perovskite oxides has been done [21–24], studying their catalytic activity, stability,



and properties both doped and undoped. Recently, La-based perovskites have been a main focus; for instance, LaCoO₃ and LaMnO₃ (and derived catalysts) have been investigated intensively for their utilization in CO oxidation [25–27], showing doping to be generally beneficial.

In this study, the previous work (see above) on La-based perovskites is extended by a series of seven doped LnFeO₃-based (Ln = La, Nd) perovskites that have not been investigated in this detail before. The perovskites were tested as catalysts in CO oxidation experiments in order to study the effect of different doping on their catalytic performance. The results were interpreted with respect to different possible reaction mechanisms (see Section 4 for more details). All materials were A-site doped with two different Ca contents (10% and 40% Ca, respectively), while the B-site was either undoped or doped with Co or Ni. Five of the investigated materials—La_{0.6}Ca_{0.4}FeO_{3- $\delta}} (La06), La_{0.9}Ca_{0.1}FeO_{3-<math>\delta}$} (La09), Nd_{0.6}Ca_{0.4}FeO_{3- $\delta}} (Nd06), Nd_{0.9}Ca_{0.1}FeO_{3-<math>\delta}} (Nd09), and Nd_{0.6}Ca_{0.4}Fe_{0.9}CO_{0.1}O_{3-<math>\delta}} (NdCo)$ have been thoroughly characterized and used as catalysts for different types of reactions in $previous studies [2,3,28,29], while a new Ni-doped catalysts—Nd_{0.6}Ca_{0.4}Fe_{0.97}Ni_{0.03}O_{3-<math>\delta$} (NdNi)—with reduced B-site doping concentration has been synthesized additionally for this study. Commercial La_{0.6}Sr_{0.4}FeO_{3- $\delta}} (LSF) was used as reference material.</sub>$ </sub></sub></sub></sub>

2. Materials and Methods

2.1. Synthesis of Ni-Doped Perovskite

All samples were prepared via the Pechini route [30]. In short, all precursors for the respective material are dissolved in doubly distilled water and HNO₃ (doubly distilled, 65%, Merck, Darmstadt, Germany). The following precursors were used for the preparation of the newly made material: Nd₂O₃ (99.9%, Strategic Elements, Deggendorf, Germany), CaCO₃ (99.95%, Sigma-Aldrich, St. Louis, MO, USA), Fe (99.5%, Sigma-Aldrich, St. Louis, MO, USA), and Ni (NO₃)₂ 6 H₂O (98%, Sigma-Aldrich, St. Louis, MO, USA). Proper ratios of the solutions are then mixed in a quartz glass flask. To facilitate the formation of cation complexes citric acid (trace metal pure, 99.9998%, Fluka, Honeywell International, Charlotte, NC, USA) is added in excess of 20%. The solvents are evaporated off and the sponge-like intermediate is heated until self-ignition occurs, after which the material is calcined for 5 h at 800 °C. The final product is then ground to a powder for subsequent experiments. Details on the synthesis can be found in Lindenthal et al. [3].

Since the initial characterization revealed the presence of an additional oxygen deficient phase, the newly synthesized material underwent an additional heat treatment step (900 °C in pure oxygen).

2.2. Sample Characterization

The newly produced perovskite was characterized thoroughly: (i) Powder X-ray Diffraction (XRD) measurements were performed to verify phase purity. (ii) Scanning Electron Microscope (SEM) images were recorded to investigate the surface topography of the material. (iii) The specific surface area according to the Brunauer–Emmett–Teller (BET) model [31] was determined to be able to compare the catalytic performance of different samples.

XRD measurements were conducted at room temperature with a PANalytical X'Pert Pro Diffractometer (Malvern Panalytical, Malvern, UK) in Bragg–Brentano geometry, equipped with an X'Celerator linear detector (Malvern Panalytical, Malvern, UK) and a mirror to separate the Cu-K_{α 1,2} radiation. The HighScore Plus software (Malvern Panalytical, Malvern, UK) [32] together with the PDF-4+2021 Database (ICDD-International Centre for Diffraction Data, Newtown Square, PA, USA) [33] were used for data analysis.

Room temperature SEM images were recorded on a Quanta 250 FEGSEM microscope (FEI Company, Hillsboro, OR, USA) using secondary electrons. All images were recorded using an acceleration voltage of 5 kV.

BET areas were calculated from adsorption-desorption isotherms that were in turn determined using an ASAP 2020 PLUS adsorption analyzer (Micromeritics Instrument

Corporation, Norcross, GA, USA). After a degassing step (for 4 h under vacuum at 300 °C), the samples were cooled to -196 °C (liquid N₂), where full N₂ adsorption-desorption isotherms were measured.

2.3. Catalytic Tests

The catalytic tests were performed in a tubular flow reactor: 10 to 20 mg of the sample were placed in a quartz glass tube (inner diameter 4 mm), mounted in an oven, and exposed to the reaction atmosphere. A total flow of 12 mL min⁻¹ was used for all catalytic experiments (4 mL min⁻¹ CO, 2 mL min⁻¹ O₂, and 6 mL min⁻¹ Ar as carrier gas, all gases purchased from the Messer Group GmbH, Bad Soden, Germany). Two different temperature programs were used: (i) 150 to 500 °C (at a rate of 1 °C min⁻¹) for all samples during the heating experiments and (ii) a temperature between 500 and 350 °C (depending on the sample, shown are curves starting at 400 °C) to 50 °C (at a rate of -1 °C min⁻¹).

Reaction progress was monitored via continuous sampling (approximately 4 minintervals) of the gas composition after the reactor using a Micro Gas Chromatograph (Micro-GC, Fusion 3000A, Inficon, Bad Ragaz, Switzerland).

Prior to catalytic testing, a blank measurement (same experimental parameters, but without catalyst) was carried out to check for effects of the reactor setup. The found amount of CO_2 was below 0.6 mol-% CO_2 at 600 °C. Therefore, the contribution of the reactor to the catalytic activity is miniscule enough that no additional corrections were performed.

To achieve comparability between the catalytic results of different samples, specific activities normalized to the sample masses and the BET areas of the respective material were used (for details refer to the Supplementary Materials). This is necessary since commonly used Turn-Over Frequencies (TOFs) require detailed information about active sites. However, due to the dynamic nature of perovskites, these details are not feasibly obtainable: For instance, vacancy concentrations in samples are highly temperature dependent, and the surfaces exposed to the reaction atmospheres in the used powder samples are highly variable as well (in terms of crystallographic orientation, termination, and actual dopant content).

3. Results

3.1. Characterization of Newly Synthesized Perovskite

XRD measurements after the synthesis of NdNi (Figure 1) revealed that the final product is not entirely phase-pure: aside from the predominant perovskite phase Nd_{0.6}Ca_{0.4}Fe_{0.97} Ni_{0.03}O_{3- δ}, small amounts of an ordered oxygen vacancy structure (Brownmillerite, Nd_{2-x}Ca_xFe_{2-v}Ni_vO₅) are also present.

In order to achieve oxygen saturation, a second heating treatment in a pure oxygen atmosphere was performed on a part of the powder; however, the second Brownmillerite phase was present even after this step. The newly synthesized perovskite was still included in the catalytic tests to also investigate the CO oxidation capabilities of Ni-doped samples (material without the second heating treatment was used for catalytic tests).

SEM images recorded to characterize the sample surface show typical characteristics of preparation by calcination and grinding: varying grain sizes resulting from the grinding process and fused small crystallites making up the overall structures are visible (for the images refer to Section 3.4, where the SEM images of the pristine catalyst are compared to the catalyst after catalytic tests).



Figure 1. XRD pattern of newly synthesized NdNi. The black stars indicate additional reflexes of an emerging impurity phase—Brownmillerite, $Nd_xCa_{2-x}Fe_{2-y}Ni_yO_5$ —which exhibits an ordered vacancy structure. All other reflexes correspond to an orthorhombic perovskite phase (only the main reflexes were indexed for clarity—for a more comprehensive reflex assignment refer to [3]).

3.2. Catalytic Results

Table 1 summarizes the results of the catalytic tests: listed are the BET areas used to calculate the specific activities (see SI for details) and the specific activities both after heating (at 500 $^{\circ}$ C) and before extinction (i.e., a marked drop of activity) during the cooling ramps.

Sample	BET Area (m ² ·g ⁻¹)	Specific Activity at 500 °C (×10 ⁻⁵ mol·s ⁻¹ ·m ⁻²)	Specific Activity Before Extinction ($\times 10^{-5}$ mol·s ⁻¹ ·m ⁻²)
La06	$2.8^{\ 1}$	7.2	2.6
La09	$3.8^{\ 1}$	6.1	3.5
Nd06	1.5^{1}	13.6	5.9
Nd09	2.2^{1}	8.5	3.4
NdCo	1.2^{1}	12.7	4.6
NdNi	2.4	7.3	3.5
LSF	5.7 ¹	4.0	1.2

Table 1. Results of catalytic tests: Specific activities at 500 °C (heating experiments) and shortly before extinction (cooling experiments). BET areas used to calculate activities are also given.

¹ BET areas taken from a previous study [29].

The progression of the specific activities during the heating experiments (Figure 2) follows the same pattern for all investigated samples: At lower temperatures an exponential increase (at low levels) is followed by a sharp jump in activity (ignition). After this ignition, the specific activities continue to rise linearly with very similar rates for all catalysts except LSF.

In the case of the four B-site undoped perovskites, clear trends emerge. Both La-based materials exhibit activities below the Nd-based ones. Additionally, there appears to be a positive effect of Ca on the activity; after ignition, both La06 and Nd06 have higher activities than their counterparts with lower amounts of Ca over the whole temperature range. However, the difference between La06 and La09 is less pronounced than for Nd06 and Nd09.



Figure 2. Comparison of CO oxidation results during heating experiments. The specific activities of the respective perovskites are shown. LSF was added as reference (full black line); the temperature progression during the catalytic test is displayed on the second y-axis. To perform the catalytic tests, a 2:1 mixture of CO and O_2 was used for stoichiometric reasons (4 mL min⁻¹ and 2 mL min⁻¹, respectively). The temperature was increased from 150 to 500 °C at a rate of 1 °C min⁻¹. The highest activities were found for the B-site undoped sample Nd06 and the Co-doped perovskite NdCo. LSF performed worse than all investigated samples.

Moreover, it can be noted that the earlier ignition happens (cf. Section 3.3), the lower the final specific activity will be: La09 displays the earliest ignition of the four but reaches only about 50% of the maximum specific activity found for Nd06. The B-site doped samples, NdCo and NdNi, are the exception: NdCo is the catalyst that ignites the earliest, but it reaches a maximum activity in the same order of Nd06. In contrast, NdNi surprisingly exhibits the highest ignition temperature while only displaying a maximum activity comparable to La06.

Co-doping appears to be beneficial, as it lowers the ignition temperature while not impeding the activity of the sample—making CO oxidation feasible at lower temperatures as well. This is not the case for Ni-doping, which, in fact, seems detrimental to the catalyst performance, as the specific activity is drastically lowered (to about 50% of the value of Nd06) and ignition is shifted to higher temperatures. However, this might be related to the presence of the additional Brownmillerite phase.

The commercial reference catalyst LSF exhibits the lowest of all found specific activities and ignites around a similar temperature as NdCo, continuing the trend of the undoped perovskites that earlier ignition leads to smaller specific activity.

Aside from the fact that Nd-based samples exhibit higher specific activities than their La-based counterparts, similar trends could not be observed for the specific activities during cooling experiments (summarized in Figure 3).

While the rates of the linear activity increase after ignition during the heating experiments are similar for all catalysts, two different groups of samples with clearly different deactivation rates emerge: The activities of the Nd-based B-site undoped samples Nd06 and Nd09, the Co-doped perovskite NdCo, and La06 decline at similar rates. This rate is lower in the case of the remaining samples (La09, NdNi, and LSF). Moreover, those samples do not display an extinction event above 50 °C.



Figure 3. Comparison of CO oxidation results during cooling experiments. The specific activities of the respective perovskites are shown. LSF was added as reference (full black line); the temperature progression during the catalytic test is displayed on the second y-axis. To perform the catalytic tests, a 2:1 mixture of CO and O₂ was used for stoichiometric reasons (4 mL min⁻¹ and 2 mL min⁻¹, respectively). The temperature was decreased from 400 to 50 °C at a rate of -1 °C min⁻¹. The highest activities right before extinction were found for the B-site undoped sample Nd06 and the Co-doped perovskite NdCo (similar to the heating experiments). While LSF exhibits the lowest activities over the whole tested temperature range, it does not show extinction above 50 °C. The same is true for La09 and NdNi.

The Nd-based B-site undoped samples exhibit inverted behavior compared to the heating experiments (Nd06 displays higher activities over the whole temperature range and deactivates at lower temperatures than Nd09), the same is not true for the La-based analogues: Initially both La06 and La09 start at similar activities; however, the deactivation rate of La06 is larger and extinction occurs at significantly higher temperatures compared to La09.

3.3. Hysteresis of Specific Activity, Ignition and Extinction

Figures 4 and 5 show the development of the specific activities during heating and subsequent cooling ramps for Nd06 and Nd09, respectively. Both samples show a pronounced hysteresis, which is present for all perovskites tested. During cooling, all samples remain in a state of high specific activity far beyond the ignition temperature; the narrowest hysteresis is that of La06 (226 °C), the largest occurs for NdNi (>391 °C, an exact value could not be determined as no extinction happened above 50 °C).

Insets in Figures 4 and 5 show an additional effect during the heating experiments. The onset of significant oxidation around the ignition temperature leads to the release of additional heat (CO oxidation is exothermic). This, subsequently, causes a spike in the temperature as the PID controller of the heating needs time to compensate for that additional heat. The brief phase of oscillations around the preset temperature arises from initial overcorrections.



Figure 4. Hysteresis of Nd_{0.6}Ca_{0.4}FeO_{3- δ} (Nd06): Ignition (onset of significant catalytic activity) upon heating the sample (red curve) happens at 423 °C, while catalytic activity persisted far below that temperature during the cooling of the sample (blue curve). For Nd06, extinction (a significant sudden drop of specific activity) was found to happen at 91 °C. The inset shows an additional thermal effect that could be observed during heating: Due to the exothermic nature of CO oxidation, the temperature spikes around ignition as additional heat is released during the reaction—the peak below the target temperature immediately after this spike arose from the PID controller of the heating overcorrecting. The inverse effect was found around extinction (not shown).



Figure 5. Hysteresis of Nd_{0.9}Ca_{0.1}FeO_{3- δ} (Nd09): Ignition (onset of significant catalytic activity) upon heating the sample (red curve) happens at 409 °C, while catalytic activity persisted far below that temperature during the cooling of the sample (blue curve). For Nd06, extinction (a significant sudden drop of specific activity) was found to happen at 146 °C. The inset shows an additional thermal effect that could be observed during heating: Due to the exothermic nature of CO oxidation, the temperature spikes around ignition as additional heat is released during the reaction—the peak below the target temperature immediately after this spike arose from the PID controller of the heating overcorrecting. The inverse effect was found around extinction (not shown).

The ignition and extinction temperatures for all samples and LSF are summarized in Table 2, as well as specific activities before extinction and after ignition. As mentioned above (see Section 3.2), trends can be derived from the ignition temperatures: Ignition occurs earlier for La-based perovskites La06, La09, and LSF; however, those three samples display lower activities compared to the Nd-based samples except for NdNi.

Table 2. Ignition and extinction temperatures of all samples and LSF. The specific activities before extinction and after ignition are also given.

Sample	Ignition Temperature (°C)	Specific Activity Before Ignition (×10 ⁻⁵ mol·s ⁻¹ ·m ⁻²)	Specific Activity After Ignition $(\times 10^{-5} \text{mol} \cdot \text{s}^{-1} \cdot \text{m}^{-2})$	Extinction Temperature (°C)
La06	387	1.1	7.7	161
La09	334	0.8	5.2	<50
Nd06	423	3.2	12.2	94
Nd09	409	0.8	7.3	146
NdCo ¹	311	1.5	7.1	70
	322	6.7	9.6	10
NdNi	441	1.4	6.7	<50
LSF	311	0.4	3.5	<50

¹ Two distinct ignitions could be observed for this sample.

Samples with lower amounts of Ca-doping at the A-site (La09 and Nd09) ignite earlier and show lower activities than samples with a higher Ca-content (La06 and Nd06).

Additionally, NdCo exhibits a two-step ignition process. A possible reason for that might be that the processes responsible for that effect happen twice (once for the main constituent Fe³⁺ and once for the dopant Co³⁺). A similar effect would be expected for NdNi; however, no two-step ignition could be observed. This probably has to do with impurity phase Brownmillerite in this sample.

No clear trends can be formulated for the extinction temperatures found in the cooling experiments. Moreover, three samples (La09, NdNi, and LSF) do not exhibit extinction at all in the investigated temperature range.

3.4. SEM Images and XRD Diffractograms after CO Oxidation

SEM images of all samples before and after the catalytic tests were recorded to ascertain whether the catalysts' surface change during CO oxidation. This is shown exemplarily for NdNi and Nd09 in Figures 6 and 7, respectively (SEM images for all other samples can be found in the Supplementary Materials, Figures S1–S5).

In the case of NdNi, no significant changes to the surface morphology were found; however, for Nd09, sub-micron sized particles are visible after reaction (Figure 7d). These particles could be identified as CaCO₃ via XRD (see Figure 8).

XRD diffractograms of Nd09, La06, and NdNi both before and after CO oxidation are displayed in Figure 8. NdNi showed no significant change after reaction (peak position and width as well as the present phases remained unchanged). Nd09 and La06 exhibit slight changes. In the diffractogram of the former, a small amount of a new phase (CaCO₃) appears after CO oxidation. XRD patterns of Nd06 and NdCo after reaction (not shown) show small amounts of CaCO₃ as well. No new phases appear in the case of La06; however, the reflexes are slightly shifted to lower angles after reaction.



Figure 6. SEM images of NdNi before (a,b) and after a heating ramp (c,d) up to 500 °C. All images were recorded with an acceleration voltage of 5 kV. (a) Overview, pristine; (b) selected area, pristine; (c) overview, after reaction; (d) selected area, after reaction. No significant changes to the material's surface are visible.



Figure 7. SEM images of Nd09 before (**a**,**b**) and after a heating ramp (**c**,**d**) up to 500 °C. All images were recorded with an acceleration voltage of 5 kV. (**a**) Overview, pristine; (**b**) selected area, pristine; (**c**) overview, after reaction; (**d**) selected area, after reaction. After reaction, small CaCO₃ particles are visible (appearing as brighter spots on the darker perovskite surface).



Figure 8. Comparison of XRD patterns of Nd09, La06, and NdNi before (green tracks) and after (red tracks) CO oxidation. Stars denote reflexes of the impurity phase Brownmillerite, the diamond marks a reflex associated with CaCO₃. Nd09 exhibits small amounts of CaCO₃ after the reaction, while both La06 and NdNi do not show any new phases after the reaction. However, the reflexes of La06 are shifted to lower angles. No significant changes in the diffractogram of NdNi were found after CO oxidation.

4. Discussion

All investigated perovskites show a similar progression of the catalytic activity upon heating: at low temperature it increases exponentially, followed by an abrupt jump (ignition). After ignition happened, the activity grows linearly; interestingly, the rate of this growth is similar for all materials studied in this work except LSF. During cooling, an inverted behavior without any exponential decrease was observed. This is because due the hysteresis, deactivation occurs at much lower temperatures than ignition (up to almost 400 $^{\circ}$ C later), resulting in the residual activity dropping to 0 immediately.

B-site doping did not lead to increased specific activities. While in the case of NdCo the activity of the undoped perovskite Nd06 was reached and ignition was reached much earlier (NdCo is the sample that ignites the earliest), doping with Ni led to a drastically reduced activity and ignition being shifted to higher temperatures (NdNi was the last catalyst to ignite). However, it can be assumed that the presence of Brownmillerite is the reason for this decline in catalyst performance, as in earlier studies on reverse water-gas shift reactions, a catalyst with 10% Ni and no competing impurity phase present showed improved activity compared to the B-site undoped material [29].

The processes leading to ignition and the catalysis mechanism of CO oxidation over perovskites are not fully understood. The shape of the activity curves and the presence of CaCO₃ in some samples suggests that different mechanisms (schematically depicted in Figure 9) play a role at different reaction conditions:

(i) At lower temperatures, both Langmuir–Hinshelwood- or Eley–Rideal-type mechanisms are plausible as both explain the exponential increase of activity. In the case of the former, catalytic activity would be hindered by full coverage of CO—as both reactants need to adsorb. Only with increasing temperature and subsequent desorption of CO—freeing adsorption sites for O₂ adsorption—can CO oxidation occur. This dependence on CO desorption might serve as an explanation for changes of specific activity as different compositions lead to different CO desorption energies. This mechanism at lower temperatures was proposed by Wang et al. for La_{0.4}Sr_{0.6}CoO_{3- δ} [34]. Yang et al. investigated a low-temperature Langmuir–Hinshelwood mechanism promoted by oxygen vacancies [35]. An Eley–Rideal mechanism in the presence of Fe³⁺ ions

was postulated for Fe_2O_3 by Waglöhner et al. [36]. Depending on surface termination and crystallographic direction of facets exposed to the reaction atmosphere, such ions are present in all investigated samples.

- (ii) Around ignition, a change of mechanisms is likely: Wang et al. suggest a Mars-van-Krevelen (MvK) mechanism for La_{0.4}Sr_{0.6}CoO_{3-δ} at high temperatures [34] and Liu et al. put forward either a conventional or a carbonate-mediated MvK mechanism for CO oxidation over Co-doped ceria catalysts [37]. This aligns with the presence of CaCO₃ in Nd06 and Nd09. Moreover, oxygen vacancies play a crucial role in MvK mechanisms and have also be found to be important for CO oxidation using Ce-doped LaCoO₃ catalysts [25]. Increasing Ca²⁺ contents lead to a higher number of vacancies in LaFeO₃- and NdFeO₃-based perovskites, which in turn lead to easier O₂ adsorption. That can explain the generally higher catalytic activities of the B-site undoped samples with 40% Ca content [28].
- (iii) The linear increase of the catalytic activity after ignition might be related to diffusion limitation.



Figure 9. Schematic representation of CO oxidation on a perovskite surface (ideal cubic unit cell shown in the middle). During a Langmuir—Hinshelwood mechanism (left) both reactants adsorb (i.a and i.b, CO_{ads} and O_{ads} , respectively), diffuse along the surface (ii), and form CO_2 , which desorbs. The Mars-van-Krevelen (MvK, right) mechanism proceeds via two steps: (i) CO adsorbs first, reacts with lattice oxygen (O_{lat}), and desorbs as CO_2 leaving an oxygen vacancy (V_O) behind. (ii) An oxygen molecule from the gas atmosphere adsorbs, splits, and re-fills the vacancy.

Moreover, the two-step ignition of NdCo suggest involvement of the B-site species. The B-site undoped species display a single step during ignition (due to only one species— Fe³⁺—making up the B-site); in the case of NdCo on the other hand, Fe³⁺ and Co³⁺ species compose the B-sublattice. Differences of CO desorption energies between Fe- and Co-sites might cause the two steps.

The results presented in this work demonstrate the flexibility of perovskites: changes in the A-site composition or the introduction of B-site dopants influence the performance of the catalyst with respect to CO oxidation, thus facilitating a possible pathway to catalyst design. Relatively simple changes to the composition of the catalyst—such as changing the A-site cation (La³⁺ vs. Nd³⁺) and/or increasing the content of A-site dopant (10% or 40% Ca)—greatly change the resulting activity and ignition temperature of the catalyst. Pursuing this design approach was the main motivation of the present study.

Supplementary Materials: The following are available online at https://www.mdpi.com/article/10 .3390/fuels3010003/s1, Figure S1: SEM images of La06, Figure S2: SEM images of La09, Figure S3: SEM images of Nd06, Figure S4: SEM images of NdCo, Figure S5: SEM images of LSF. Author Contributions: Conceptualization, C.R.; methodology, C.R., F.S., L.L. and R.B.; validation, F.S., L.L. and T.R.; formal analysis, L.L. and T.R.; investigation, R.B. and L.L.; resources, L.L. and R.B.; data curation, L.L. and R.B.; writing—original draft preparation, R.B. and T.R.; writing—review and editing, C.R., F.S., L.L. and T.R.; visualization, F.S., L.L. and C.R.; supervision, C.R.; project administration, C.R.; funding acquisition, C.R. All authors have read and agreed to the published version of the manuscript.

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